

TEM studies of Ge nanocrystal formation in PECVD grown SiO₂:Ge/SiO₂ multilayers

S Ağan¹, A Dana² and A Aydın²

¹ Physics Department, Kırıkkale University, 71450 Kırıkkale, Turkey

² Physics Department, Bilkent University, 06800 Ankara, Turkey

Received 16 January 2006

Published 16 May 2006

Online at stacks.iop.org/JPhysCM/18/5037

Abstract

We investigate the effect of annealing on the Ge nanocrystal formation in multilayered germanosilicate–oxide films grown on Si substrates by plasma enhanced chemical vapour deposition (PECVD). The multilayered samples were annealed at temperatures ranging from 750 to 900 °C for 5 min under nitrogen atmosphere. The onset of formation of Ge nanocrystals, at 750 °C, can be observed via high resolution TEM micrographs. The diameters of Ge nanocrystals were observed to be between 5 and 14 nm. As the annealing temperature is raised to 850 °C, a second layer of Ge nanocrystals forms next to the original precipitation band, positioning itself closer to the substrate SiO₂ interface. High resolution cross section TEM images, electron diffraction and electron energy-loss spectroscopy as well as energy-dispersive x-ray analysis (EDAX) data all indicate that Ge nanocrystals are present in each layer.

1. Introduction

Recently, different techniques have been tried to fabricate quantum-size particles of semiconductors. The information about structural and morphological characteristics of these semiconductor nanocrystals is very important for modelling electronic properties and designing the structure of optoelectronic devices. The potential applications of the Ge nanocrystals have attracted much attention due to optical and electronic properties, which may be photodetectors [1], light emitters [2], single electron transistors [3], photonic structures [4], quantum computers [5], non-linear optical media [6] and photosensitive materials [7]. In particular, embedding silicon, germanium or silicon–germanium nanocrystals in an insulator matrix has been proposed for non-volatile memory devices [8]. As charge loss through lateral paths in nanocrystal based memory devices is suppressed by the oxide isolation between nanocrystals, these devices exhibit superior charge retention characteristics compared with conventional floating-gate memory devices [9–11]. Retention time in a quantum dot memory device is reported to be strongly related to the shape and the size of the nanocrystals [7]. Retention time improvement of an aligned Si double layer nanocrystal structure has also been reported [12] by scaling the lower layer nanocrystal size. Even though the stacked nanocrystals are not exactly aligned, the lower layer nanocrystals can still help to reduce charge leakage

from the upper layer nanocrystals [13]. Finally, further increases in memory density may be envisioned through multilayered structures.

The key issue, therefore, in the growth process is to synthesize Ge nanocrystals in an insulating matrix, preferably in the form of thin films, with a narrow size distribution and low defect density. For memory device applications, it is also crucial to control the thickness of the SiO₂ tunnel oxide underneath the nanocrystal layer, as well as the density and size of the Ge nanocrystals. While a number of techniques to prepare Ge nanocrystals are being used, germanosilicate films deposited by PECVD have some advantages due to low temperature of deposition, excellent step coverage characteristics, high blocking effects against moisture and alkaline ions, and relatively high dielectric constant values [14]. The direct growth process is probably the most promising among the techniques that have been proposed. PECVD is also very suitable for growth of multilayers with differing compositions. TEM is a powerful technique for the structural and chemical investigation of a wide range of materials. It provides direct evidence of phase separation and crystallization, as well as crystal size and distribution. The ability to determine the size and structure of nanoparticles would provide information to allow the fabrication of structures with desired electrical and optoelectronic properties for device applications.

TEM has, previously, been used for structural analysis of Ge nanocrystals fabricated by cosputtering and annealing [15, 16]. Cross sectional TEM images provided direct evidence of the nanocrystal formation. It has been observed that lower Ge content results in smaller Ge nanocrystal size. Similar TEM studies of Ge nanocrystals have been performed on samples prepared by the ion implantation technique [17]. With increasing annealing time, it has been observed that the mean density of Ge nanocrystals decreases while the average size of the nanocrystals increases. Choi *et al* have also reported results of TEM studies of Ge nanocrystals obtained by rapid thermal annealing of cosputtered (Ge and SiO₂) samples [18]. At 900 °C they have observed two regions with different nanocrystal densities. At higher temperatures nanocrystals were observed to form only at the Si/SiO₂ interface. We have previously studied Ge nanocrystals using TEM in single layers grown by PECVD [19] and post-annealing. At an annealing temperature of 1010 °C, we found that smaller crystals form at the Si/SiO₂ interface while larger size crystals form away from the interface.

In the literature, Si nanocrystals in Si/SiO₂ multilayers were obtained by a low-pressure chemical vapour deposition (LPCVD) of thin silicon layers and atmospheric pressure chemical vapour deposition (APCVD) of SiO₂ layers followed by high-temperature thermal oxidation [20, 21]. Molecular beam epitaxy [22] and electron beam deposition [23] have also been used. However, there is no report of Ge nanocrystal formation in SiO₂ multilayers in the literature.

In this work, we investigate the effect of annealing on the Ge nanocrystal formation in multilayered germanosilicate–oxide multilayers grown on Si substrates by plasma enhanced chemical vapour deposition (PECVD). The multilayered samples were annealed at temperatures ranging from 750 to 900 °C for 5 min under nitrogen atmosphere. The formation and size of Ge nanocrystals were determined from high resolution electron microscopic (HREM) observations. The samples for the cross-sectional HREM observations were prepared by standard procedures including mechanical and low temperature (200 K) Ar-ion milling techniques with care taken to minimize radiation damage due to impacting Ar⁺ ions.

2. Experimental procedure

The multi-layered structures, with different annealing temperatures, from 750 to 900 °C, were prepared. The oxide (10 nm)–germanosilicate (20 nm)–oxide (30 nm) multi-layered films were

grown in a PECVD reactor (model PlasmaLab 8510C) on Si substrates using 180 sccm SiH₄ (2% in N₂), 225 sccm NO₂ and varying flow rates of GeH₄ (2% in He) as precursor gases, at a sample temperature of 350 °C, a process pressure of 1000 mTorr under and an applied RF power of 10 W. First, a tunnel oxide followed by a deposition of a SiO₂:Ge layer and a layer of SiO₂ layer was grown. SiO₂:Ge and SiO₂ layers form a pair. Samples with layers ranging from three to 10 pairs have been grown. The germanium doped SiO₂:Ge layer is where nanocrystals form upon annealing. Films were grown on p-type silicon substrates with resistivity of 55 Ω cm. The samples were then annealed in N₂ atmosphere in a quartz oven at temperatures ranging from 750 to 900 °C for 5 min. The samples were loaded and unloaded in ramp times of 1 min.

Samples for TEM observation were prepared in cross-section orientation, so that the film layers were viewed edge-on. This preserves the information about the position of the nanocrystals with respect to the surface. The structural characterization was carried out in a JEOL 2010F field-emission transmission electron microscope at 200 kV making use of diffraction, tilt and rotation facilities.

3. Results and discussion

TEM studies of several samples were carried out at 200 kV. After a brief survey of Ge precipitation, several clusters in each sample were studied for structural detail using high resolution selective area diffraction. High resolution scanning TEM images are formed in the JEOL 2010F by focusing coherent electrons into a probe and scanning mode across the sample.

Most of the data presented in this work are obtained on samples with a composition of Si_{1.0}Ge_{0.54}O_{3.4} as determined by x-ray photoelectron spectroscopy where the germane flow rate during deposition was 90 sccm. Multilayers of up to 10 pairs have been grown. The multilayered samples were later annealed at temperatures ranging from 750 to 900 °C for 5 min in nitrogen atmosphere. TEM observations indicate that the Ge nanocrystals can be arranged at a specific depth and for a specific number of layers with nanometre precision.

In figure 1(a), bright field images of samples with three pairs of SiO₂/SiO₂:Ge layers annealed at 750 °C obtained via TEM are shown as an example. We observe the formation of three dark bands separated by light bands of oxide, indicating diffusion and precipitation of Ge atom clusters in each Ge doped SiO₂ layer. It seems from figure 1(a) that the initial growth rates of both the oxide layer and the germanium doped layer are smaller than the following layers. A high resolution image of the same sample is shown in figure 1(b). We find that crystallization of Ge clusters has started and crystal planes are observable. Crystallized Ge forms a quasi-continuous layer with crystal planes interwoven with defects. Crystal sizes are on the order of 5 nm. The amorphous character of the surrounding SiO₂ matrix as well as the crystal planes of the silicon substrate is also observable. We have taken electron diffraction data from the same area of the sample, shown in figure 1(c). Diffraction data indicate that there are a lot of defects in the Ge layers. We find that these samples are at the onset of crystallization for the Ge clusters. Similar TEM data of other samples annealed at 800, 825, 850 and 900 °C show that as the anneal temperature increases the defect density in the nanocrystals decreases and the size of the nanocrystals increases. Figure 2 illustrates the progress of crystallization further. The sample in figure 2 is annealed at 825 °C. We see a combination of crystallized and amorphous germanium clusters. The average sizes of the crystallized Ge are 8 nm. We clearly observe a twin boundary in the lower Ge nanocrystal. However, the Ge nanocrystal in the upper part of the micrograph shows perfect crystallization without any defects. Observation of perfect crystal planes for the Si substrate shows the high quality of the TEM images. A small cluster in between two nanocrystals does not show crystallization. This indicates that all Ge clusters are not crystallized at this annealing temperature and time. We also note the absence of

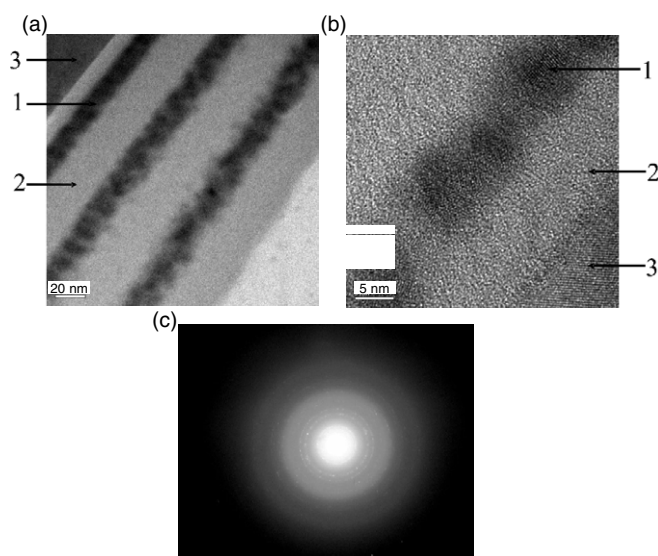


Figure 1. Bright field TEM micrograph of a three pair sample annealed at 750 °C for 5 min. The samples were grown with 90 sccm of germane flow rate. Ge clusters are observed as dark bands separated by SiO₂ layers (a) and exhibit crystal planes confirming the formation of nanocrystals when studied under high resolution (b). In (a) we point to Ge clusters (1), SiO₂ spacer layers (2), and Si substrate (3). The electron diffraction pattern of the same sample exhibits hazy rings superimposed on a diffraction pattern indicating the presence of a high density of defects.

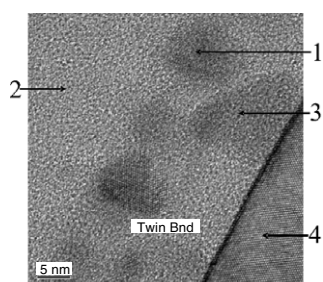


Figure 2. Cross section TEM micrograph of a sample annealed at 825 °C for 5 min. The samples were grown with 90 sccm of germane flow rate. Twin boundary defects in Ge nanocrystals (1), the oxide matrix (2), Cu contamination from the sample grid (3) and the Si substrate (4) are indicated.

nanocrystals in the near interface for a narrow band of oxide layer. The dark irregular patch (3) in figure 2 is Cu contamination that has been identified by EDAX analysis near the Si substrate (4) interface. This may be due to the Cu grid used to hold the sample during the ion beam milling process.

In order to study the influence of Ge concentration in the as grown material on the crystallization of Ge nanocrystals, we have also prepared samples grown with germane flow rate of 120 sccm. These films have a composition of Si_{1.0}Ge_{0.67}O_{3.6}. The high resolution TEM micrograph of these samples is shown in figure 3. The oxide matrix as well as a continuous band of Ge clusters is clearly observed. The Ge nanocrystal planes are well resolved. Due to excess Ge all the nanocrystals are connected. A number of twin boundaries separate various

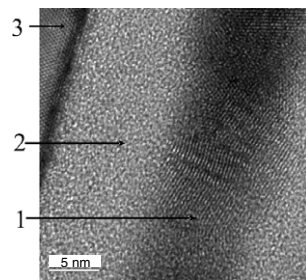


Figure 3. Cross section TEM micrograph of a sample grown with 120 sccm of germane flow rate and annealed at 800 °C for 5 min. Higher Ge concentration results in continuous distribution of Ge crystal planes (1) with locally changing planar directions surrounded by the oxide matrix (2) and the Si substrate (3).

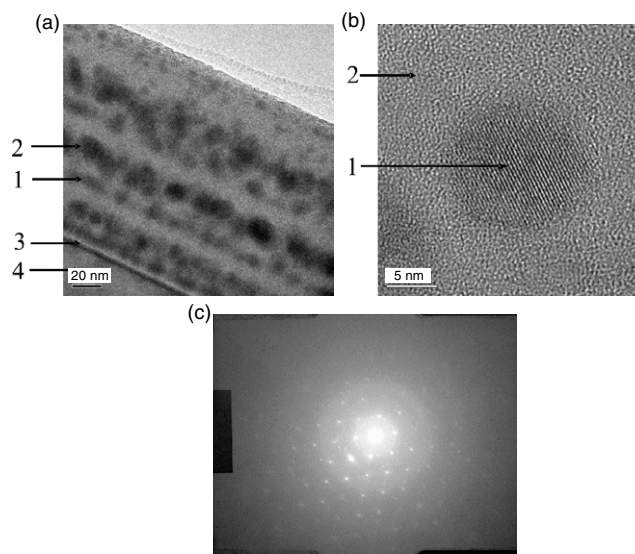


Figure 4. Bright field TEM cross section micrographs with Ge nanocrystals for samples grown with 90 sccm Ge flow rate and annealed at 900 °C for 5 min. Small (1) and large Ge nanocrystals (2) as well as the thin layer of oxide at the interface (3) are indicated. High resolution TEM study of the same sample shows perfectly aligned crystal planes surrounded by the oxide matrix. Electron beam diffraction patterns confirm excellent crystallinity of the same nanocrystal (c).

nanocrystals from each other. Ge nanocrystal layers are separated from each other with SiO₂ layers. We feel that multiple buried epitaxial Ge crystal layers may be obtained from these samples grown with high Ge concentration in the oxide matrix, under optimized processing conditions.

As an example of full crystallization with no observable defects, we show a bright field TEM micrograph of the sample grown with 90 sccm of germane flow rate and annealed at 900 °C in figure 4. The sample contains three pairs of SiO₂:Ge/SiO₂ layers. Dark patches are Ge nanocrystals in the surrounding SiO₂ matrix. It is clearly seen that as the annealing temperature increases, two distinct layers of Ge nanocrystals form in each Ge doped layer. One of the layers is composed of relatively large nanocrystals, whereas the other layer is composed

of smaller size nanocrystals. The bilayer formation starts above 850 °C in the samples studied. The smaller size nanocrystals are formed closer to Si substrate followed by larger nanocrystals. TEM micrographs suggest that the smaller Ge nanocrystals evolve from the larger nanocrystals and diffuse and expand into the oxide layers separating the germanosilicate layers (figure 4(a)).

A high resolution TEM image of a typical nanocrystal from the same sample annealed at 900 °C for 5 min is shown in figure 4(b). This illustrates an example of full crystallization in a nanocrystal size of 14 nm in diameter with no observable defects. High resolution micrographs and selective area diffraction confirm that perfect Ge nanocrystals are formed in these samples. It can clearly be seen that Ge nanocrystals have well defined spherical shapes. The size of the crystal islands can be determined from the TEM images. It should be noted that Ge nanocrystal size increases with increasing annealing temperatures. The micrograph shows perfect alignment of crystallographic planes in the Ge nanocrystal that has clearly pronounced facets. The average separation between individual Ge nanocrystals is about 4 nm, corresponding to the nanocrystal density.

In figure 4(c), the crystallinity of the islands was further confirmed by selective area electron diffraction of Ge nanocrystals for a representative sample. We have obtained excellent diffraction patterns compared with those obtained from the samples annealed at 750 °C. This indicates that at 900 °C full crystallization of Ge clusters takes place. These Ge nanocrystals do not exhibit any defects. It is interesting to note that the crystal quality of the sample annealed at 900 °C for 5 min is the same as those obtained at 1010 °C annealed for 1 h. This suggests significant savings in the thermal budget of future devices employing these nanocrystals [18]. In most embedded Ge nanocrystal samples, the interplane distances measured from HRTEM images have led us to conclude that the nanostructures possess diamond structure. Due to the electron diffraction patterns, one could say that the Ge nanocrystals of figure 4(c) show typical crystal habits of a diamond structure; these images are very close to the projection of a cuboctahedron, which consists of eight (111) planes and six (100) planes, from the [100] direction.

We have performed a detailed analysis of TEM micrographs for various samples in order to extract information on the diffusion characteristics of Ge in the SiO₂ matrix. The result of this analysis from a two pair sample is displayed in figure 5, where Ge concentration as a function of distance from the Si substrate interface is shown. The figure demonstrates the precipitation of Ge in layers as a function of the annealing temperature in the ranges of 750–900 °C. At low temperatures Ge precipitation occurs in the middle of the germanosilicate layers and no bilayer formation is seen. As the annealing temperature increases, broadening of the Ge profiles is observed. At 850 °C, broadening is accompanied by a second hump on each Ge peak on the Si substrate interface side. At 900 °C, a clear shift of the Ge profile towards the Si substrate interface is accompanied by distinctly observable secondary precipitates closer to the interface. A comparison of the total area under the Ge distribution curves for the sample annealed at 750 and 900 °C indicates that the amount of Ge that precipitated at 750 °C is not conserved as the annealing temperature is increased. Ge still remaining in the SiO₂:Ge matrix continues to precipitate, accompanied, possibly, by out-diffusion of Ge from the larger size Ge nanocrystals. At this point it is not clear why the additional precipitating band of Ge at higher temperatures does not merge with the initial Ge band broadening it. One possibility is the release of Ge atoms from GeO_x environments with Si atoms out-diffusing from the substrate [24]. This may explain the precipitation of the second smaller Ge nanocrystal layer closer to the Si substrate interface. However, our observations of the Si substrate crystal planes at the interface do not provide evidence for this possibility. Alternatively, diffusion of Ge atoms aided by increasing stress fields towards the interface may be considered.

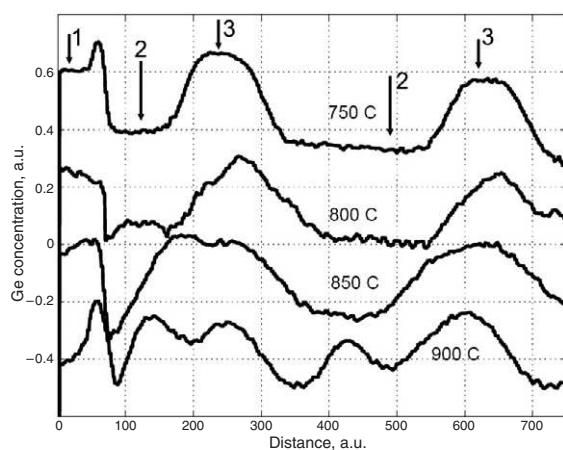


Figure 5. Assuming that the Ge concentration is proportional to the grey scale of the TEM micrograph, Ge concentration profiles have been calculated for several samples grown with 90 sccm germane flow rate and annealed at 750–900 °C for 5 min. The evolution of the smaller size Ge nanocrystals forming a bilayer is clearly observed at 850 and 900 °C.

The observation of the formation of PECVD grown high quality Ge nanocrystals in multilayers of Ge doped oxide layers is an important result of this work. The separation of Ge nanocrystals with different sizes into two layers is also a crucial result of this study and may be especially important for flash memory applications. Both results suggest that it may be possible to grow multilayered stacks of Ge nanocrystals with controlled separations ranging from a few nanometres to tens of nanometres which may help improve retention times in nanocrystal memory devices. A long retention time will be possible in the doubly stacked nanocrystal memory, since the charge leak between the upper nanocrystals and the channel can be suppressed by an energy barrier due to quantum confinement and Coulomb blockade in the lower nanocrystals. Charge retention will be improved by making the lower nanocrystal size smaller, because the energy barrier is higher in smaller nanocrystals. From electrical measurements it is found that the retention time of the double layer nanocrystal devices over the single layer ones is much improved. The charge transfer prefers direct tunnelling through thin tunnelling and interlayer oxides even in the double layer nanocrystal structure [14]. Further investigation of the dynamics of this bilayer formation may shed light into ways of controlling the size of the nanocrystals.

We have also grown 10 pairs of SiO₂/SiO₂:Ge layers and each one has been separated with about 20 nm oxide thicknesses. These samples have been grown with germane flow rates of 90 sccm followed by annealing at 800 °C for 4 min. Cross section TEM image in figure 6(a) shows 10 germanosilicate layers with nanocrystal formation. Here, germanosilicate layers (1), oxide layers (2) and Si substrate (3) are identified. Each period consists of a 20 nm germanosilicate layer and 40 nm SiO₂. We note that bilayer formation for these samples was not observed since the anneal time (4 min) is short and anneal temperature is relatively low (800 °C). Based on an HRTEM image of these samples, figure 6(b) shows Ge nanocrystals (1) with an average size of 8 nm. The micrograph shows perfect alignment of crystallographic planes in the Ge nanocrystals that have clearly pronounced facets. A closer look at different nanocrystals shows that different orientations of crystallographic planes in the Ge nanocrystals are present, indicating the presence of twin boundaries. We conclude that Ge nanocrystals form in each layer on the multilayer sample.

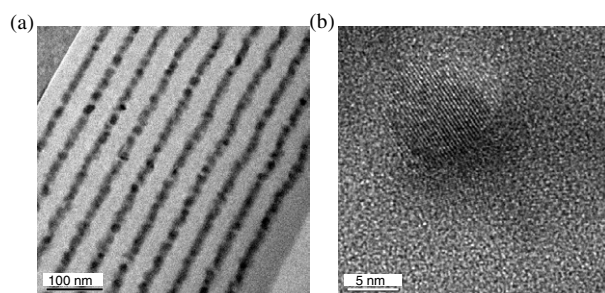


Figure 6. Cross section TEM micrograph of a 10 pair multilayered sample, grown with 90 sccm of germane flow rate annealed at 800 °C for 4 min exhibiting Ge nanocrystal bands (a), and a high resolution TEM image of the Ge nanocrystals (b). Note that Ge nanocrystal bilayers have not formed due to low temperature of annealing.

4. Conclusions

Using HRTEM, EDAX, and electron diffraction we have shown the formation of crystalline Ge nanocrystals in SiO₂ matrices grown with the PECVD technique. Each layer in multilayers of SiO₂:Ge/SiO₂ contains Ge nanocrystals. The mean size of the Ge nanocrystals was found to be between 5 and 14 nm depending on the annealing temperature and duration. With increasing annealing temperature, defect density decreases and crystal sizes increase. At 900 °C, a bilayer of Ge nanocrystals forms with a band of small size Ge nanocrystals and a band of larger size nanocrystals. These Ge nanocrystals embedded in the SiO₂ multilayers have potential for application in memory devices. We have found that annealing temperature is a critical parameter for the formation of Ge nanocrystals. The temperature and time duration of annealing determine the size and quality of Ge nanocrystals. Multilayered stacks of Ge nanocrystals with separations ranging from a few nanometres to tens of nanometres can be obtained suitable for future memory devices.

Acknowledgments

This work is supported by SEMINANO, a European Union FP6 project and by a Scientific and Technical Research Council of Turkey (TUBITAK) grant, TBAG-85/U. TEM work was carried out at the Center for Microanalysis of Materials, University of Illinois, which is partially supported by the US Department of Energy (DEFG02-91-ER45439). One of us (SA) gratefully acknowledges the financial support of the Scientific and Technical Research Council of Turkey (TUBITAK) to visit UIUC. We thank Dr A Çelik-Aktaş for help with TEM work and Professor J M Zuo for his hospitality. We gratefully acknowledge Professor S Suzer of Bilkent University Chemistry Department for the XPS measurements.

References

- [1] Wang K L, Liu J L and Jin G 2002 *J. Cryst. Growth* **237–239** 1892
- [2] Wang Y Q, Kong G L, Chen W D, Diao H W, Chen C Y, Zhang S B and Liao X B 2002 *Appl. Phys. Lett.* **81** 4174
- [3] Averin D V and Likharev K K 1986 *J. Low-Temp. Phys.* **77** 2394
- [4] Meseguer F, Blanco A, Miguez H, Garcia-Santamaria F, Ibasate M and Lopez C 2002 *Colloids Surf.* **202** 281
- [5] Shlimak I, Vagner I and Safarov V I 2000 *Proc. 25th Int. Conf. on the Physics of Semiconductors* (Osaka: Springer)
- [6] Jie Y E, Xiong Y N, Wee A T S, Huan C H A and Ji W 2000 *Appl. Phys. Lett.* **77** 3936

- [7] Nishii J, Kintaka K, Hosono H, Kawazoe H, Kato M and Muta K 1999 *Phys. Rev. B* **60** 7166
- [8] Kim D W, Prins F E, Kim T, Hwang S, Lee C H, Kwong D L and Banerjee S K 2003 *IEEE Trans. Electron Devices* **50** 510
- [9] Baron T, Pelissier B, Perniola L, Mazen F, Hartmann J M and Rolland G 2003 *Appl. Phys. Lett.* **83** 1444
- [10] De Blauwe J 2002 *IEEE Trans. Nanotechnol.* **1** 72
- [11] Niquet Y, Delerue C, Allan G and Lannoo M 2000 *The Fall 2000 MRS Mtg (Boston, MA)*
- [12] Ohba R, Sugiyama N and Uchida K 2002 *IEEE Trans. Electron Devices* **49** 1392
- [13] Lee C, Gorur-Seetharam A and Kan E C 2003 *IEDM Tech. Dig.* 557
- [14] Maeda M, Yamamoto E, Ohfuji S and Itsumi M 1999 *J. Vac. Sci. Technol. B* **17** 201
- [15] Serincan U, Kartopu G, Guenness A, Finstad T G, Turan R, Ekinci Y and Bayliss S C 2004 *Semicond. Sci. Technol.* **19** 247
- [16] Kolobov A V, Wei S Q, Yan W S, Oyanagi H, Maeda Y and Tanaka K 2003 *Phys. Rev. B* **67** 195314
- [17] Bonafos C, Garrido B, Lopes M, Perez-Rodriguez A, Morante J R, Kihn Y, Ben Assayag G and Claverie A 2000 *Appl. Phys. Lett.* **76** 3962
- [18] Choi W K, Ho V, Ng V, Ho Y W, Ng S P and Chim W K 2005 *Appl. Phys. Lett.* **86** 143114
- [19] Ağan S, Çelik-Aktaş A, Zuo J M, Dana A and Aydınlı A 2006 *Appl. Phys. A* **83** 107
- [20] Modreanu M, Aperathitis E, Androulidaki M, Audier M and Chaix-Pluchery O 2005 *Opt. Mater.* **27** 1020
- [21] Photopoulos P, Nassiopoulou A G, Kouvatso D N and Travlos A 2000 *Mater. Sci. Eng. B* **69/70** 345
- [22] Lockwood D J, Lu Z H and Baribeau J M 1996 *Phys. Rev. Lett.* **76** 539
- [23] Nihonyanagi S, Nishimoto K and Kanemitsu Y 2002 *J. Non-Cryst. Solids* **299–302** 1095
- [24] Maeda Y 1995 *Phys. Rev. B* **51** 1658

Corrigendum

TEM studies of nanocrystal formation in PECVD grown for SiO₂:Ge/SiO₂ multilayers

S Agan 2006 *J. Phys.: Condens. Matter* **18** 5037–5045

In figure 4(c) the diffraction pattern of the Si substrate was inadvertently included instead of the diffraction pattern of Ge nanocrystals, as shown below:

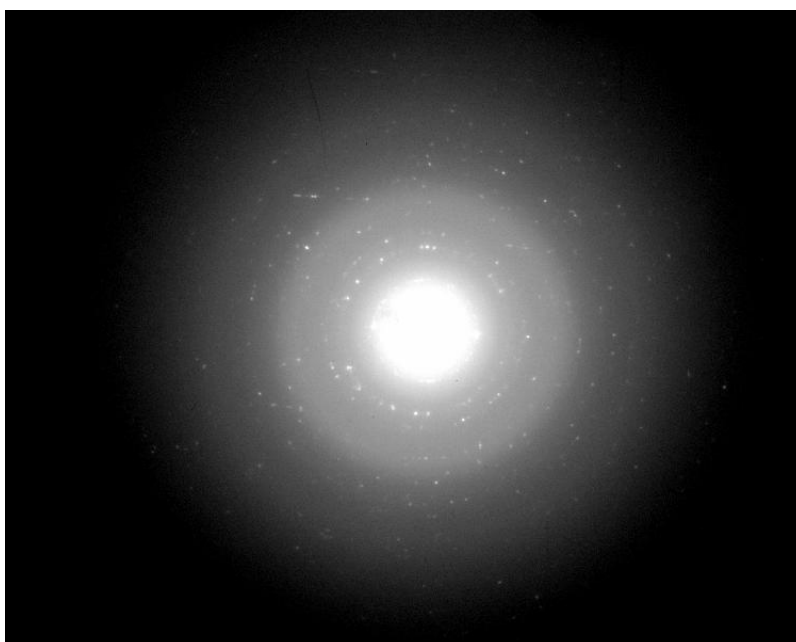


Figure 4(c). Electron beam diffraction pattern of Ge nanocrystals